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# The Crystal Structure of Chromium Thiophosphate, CrPS<sub>4</sub>

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Chromium thiophosphate, CrPS<sub>4</sub>, is monoclinic with a = 10.871 (2), b = 7.254 (2), c = 6.140 (2) Å,  $\beta = 91.88$  (4)°, V = 483.9 (4) Å<sup>3</sup>, space group C2 and Z = 4. The crystal structure has been determined by Fourier methods from the X-ray intensities of 911 independent reflexions measured on a four-circle automatic single-crystal diffractometer, and refined to R = 0.019. It is characterized by puckered hexagonally close-packed sulphur layers stacked parallel to (100) with Cr in octahedral and P in tetrahedral interstices. CrPS<sub>4</sub> is the first metal thiophosphate with the metal atom in octahedral sulphur coordination.

## Introduction

Metal phosphorus sulphides of compositional type  $M_x PS_4$  (x values so far known are 1.0, 1.5, 3.0) must be looked upon as thiophosphates. The determinations of the crystal structures of BPS<sub>4</sub> (Weiss & Schaefer, 1963), AlPS<sub>4</sub> (Weiss & Schaefer, 1960), GaPS<sub>4</sub> (Buck & Carpentier, 1973), InPS<sub>4</sub> (Carpentier, Diehl & Nitsche, 1970), BiPS<sub>4</sub> (Zimmerman, Carpentier & Nitsche, 1975), Pd<sub>3</sub>(PS<sub>4</sub>)<sub>2</sub> (Bither, Donohue & Young, 1971) and Cu<sub>3</sub>PS<sub>4</sub> (Ferrari & Cavalca, 1948) have shown that the tetrahedral PS<sub>4</sub> group is a common feature of all  $M_x PS_4$  compounds so far studied. This class of compounds has gained interest not only because of its manifold crystal chemistry but also for potential applications in solid-state devices since many of the compounds crystallize in non-centrosymmetric space groups.

A new metal thiophosphate with M = Cr and x = 1.0 has recently been synthesized. To obtain further information on the crystal chemistry of thiophosphates the crystal structure of CrPS<sub>4</sub> has been investigated by X-ray diffraction methods.

## Experimental

 $CrPS_4$  was synthesized from powdered stoichiometric amounts of the elements in evacuated quartz ampoules at a temperature of 700 °C. Reaction velocity was enhanced by adding 5 mg of iodine per cm<sup>3</sup> of the tube volume. Black, lustrous single crystals were obtained by chemical transport in a temperature gradient from

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750 to  $650^{\circ}$ C. The very soft and plastic crystals were of ribbon-like or prismatic habit, slipping very easily parallel to the dominant crystal face, (001).

1400

From Weissenberg photographs the crystals were found to be monoclinic. Systematic extinctions hkl with h + k = 2n + 1 indicated the possible space groups to be C2/m, C2, or Cm. A piezo test could not be performed because of the high conductivity of the crystal samples.

Precise lattice constants were obtained from Guinier powder photographs (Cu  $K\alpha$  radiation,  $\lambda = 1.54178$  Å) with cubic As<sub>2</sub>O<sub>3</sub> (a = 11.081 Å) as an internal standard. It followed from an indexing of the crystal faces that the crystal form is built up by a combination of {100}, {001}, { $\overline{101}$ }, {201} and {120}. {001} is the dominant crystal form.

X-ray fluorescence analysis with InPS<sub>4</sub> and Cr<sub>2</sub>S<sub>3</sub> as standards yielded atomic ratios P:S and Cr:S both of 1:4. The chemical formula was thus confirmed to be CrPS<sub>4</sub>. By the suspension method with bromoformmethanol, a density  $\rho_m = 2.88$  (3) g cm<sup>-3</sup>, was obtained resulting in four formulae per unit cell (Z = 4). All crystal data are summarized in Table 1.

For X-ray intensity measurements a prismatic crystal specimen of  $0.32 \times 0.27 \times 0.24$  mm was used. Intensity data of 911 independent reflexions in the range  $0.0 \leq (\sin \theta)/\lambda \leq 0.75$  Å<sup>-1</sup> were measured with a NaI(Tl) scintillation counter on a four-circle automatic single-crystal diffractometer (Enraf–Nonius CAD-4) using Zr-filtered Mo Ka radiation ( $\lambda = 0.71056$  Å) monochromatized by a graphite monochromator. The background-corrected intensities were further corrected for absorption ( $\mu = 14.6$  cm<sup>-1</sup>) with the program *ORABS* (Busing, 1966). Corrections for long-time drift (standard reflexions: 211 and 421), Lorentz and

## Table 1. Crystal data of $CrPS_4$

a = 10.871(2) Å	$\rho_m = 2.88$ (3) g cm <sup>-3</sup>
b = 7.254(2)	$\rho_x = 2.898$
c = 6.140(2)	Z = 4
$\beta = 91.88 (4)^{\circ}$	FW 211-2258
$V = 483 \cdot 9 (4) \text{ Å}^3$	F(000) = 412  e

polarization factors were performed with the X-RAY 70 system (Stewart, Kundell & Badwin, 1970), which was also used for structure determination and refinement.

#### Determination and refinement of the crystal structure

A three-dimensional Patterson synthesis was calculated from all  $F_o^2$  values giving a map with very sharp peaks on an extremely low background. Although the Wilson plot indicated centrosymmetry a complete interpretation of the Patterson map resulted in an atomic arrangement which was inconsistent with the centrosymmetric space group C2/m. Therefore, two Patterson-compatible structural models were established, one in space group C2, the other in Cm [No. 5 and No. 8, respectively, in *International Tables for X-ray Crystallography* (1965)].

For structure factor calculations, atomic scattering factors for Cr, P and S (neutral atoms) were taken from Cromer & Waber (1965).  $\Delta f'$  and  $\Delta f''$  correction factors for anomalous dispersion (Cromer & Liberman, 1970) were also included. A least-squares refinement was performed with the program *CRYLSQ* of the X-RAY 70 system (Stewart *et al.*, 1970). The function to be minimized was  $\Sigma w ||F_o| - |F_c||^2$  with  $w = 1/\sigma^2$  ( $F_o$ ),  $\sigma$  being the estimated standard deviation (e.s.d.) which was derived from counting statistics. 101 reflexions having  $F_o < 3\sigma$  were treated as 'less-thans'. Refinement was started with positional parameters as derived from the Patterson map, and individual isotropic temperature factors.

Refinement in space group C2 yielded a reliability index  $R = \Sigma ||F_o| - |F_c||/\Sigma F_o$  of 0.085. For space group Cm, R = 0.104 was obtained. Hence, the C2 model was selected for further refinement. With individual anisotropic temperature coefficients (Cruickshank, 1956), dispersion correction factors, and an isotropic correction factor for secondary extinction (Larson, 1967) the reliability index converged to R =0.019 with 'less-thans' omitted and R = 0.022 with 'less-thans' included. Different weighting schemes were

Table 2. Atomic positional parameters and anisotropic thermal coefficients  $(\times 10^4)$ 

	Cr(1)	Cr(2)	Р	S(1)	S(2)	<b>S</b> (3)	S(4)
Position	2( <i>a</i> )	2( <i>a</i> )	<b>4(</b> <i>c</i> <b>)</b>	4( <i>c</i> )	4( <i>c</i> )	<b>4</b> ( <i>c</i> )	4( <i>c</i> )
x	0	0	2971 (1)	1335 (3)	1347 (3)	1062(1)	1303(1)
у	0	5095 (2)	2588 (6)	-84(6)	5248 (6)	2594 (5)	7575(5)
Ζ	0	0	1655(1)	7007 (5)	6993 (5)	1934 (1)	1452 (1)
$U_{11}$	65 (10)	71 (11)	64 (03)	101 (13)	134 (14)	64 (03)	80 (03)
$U_{22}^{''}$	86 (09)	82 (09)	100 (03)	141 (09)	120 (12)	105 (03)	110(03)
U.,	148 (11)	104 (10)	118 (03)	161 (14)	142 (13)	132 (03)	108 (03)
$U_{12}^{*}$	0	0	17(12)	44 (10)	-15(10)	32(13)	16(13)
$U_{13}^{\prime \prime}$	18 (08)	00 (08)	02 (02)	20(09)	36 (09)	03(02)	04(02)
$U_{23}^{(1)}$	0	0	30(11)	49 (09)	-27 (09)	-22(13)	-17(12)

applied to the observed structure factors. Best agreement between the sets of  $F_o$  and  $F_c$  was obtained with unit weights.\* The final atomic positional and anisotropic thermal parameters are summarized in Table 2.

A difference Fourier map was completely featureless, the highest 'peak' being about 1% of the smallest peak of the  $F_o$  synthesis.

#### Discussion

As is shown on Fig. 1 viewed normal to (001), the structure of  $CrPS_4$  is characterized by puckered S layers arranged in hexagonal close-packing parallel to (100), the stacking sequence being shown on the left-hand side of the figure. Both symmetrically inequivalent Cr atoms are coordinated each by six S atoms in the form of a slightly distorted octahedron. The P atoms are located in the centres of rather regular S tetrahedra. One of the characteristic coordination polyhedra around Cr and P is outlined.

Bonding distances and angles of the pertinent coor-



Fig. 1. Layer-like atomic arrangement of the  $CrPS_4$  crystal structure viewed along  $[00\overline{1}]$ , with stacking sequence.

<sup>\*</sup> A list of structure factors has been deposited with the British Library Lending Division as Supplementary Publication No. SUP 32253 (5 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 13 White Friars, Chester CH1 1NZ, England.

Table 3. Bond distances (Å) and angles (°)

Coordinates of bonding atoms (see Fig. 1)

	$\begin{array}{ccc} Cr(1^{i}) & x_{ij} \\ Cr(1^{ii}) & \frac{1}{2} + \\ Cr(2^{i}) & x_{ij} \\ Cr(2^{ii}) & \frac{1}{2} + \\ P^{i} & x_{ij} \\ S(1^{i}) & x_{ij} \end{array}$	$\begin{array}{l} y, z \\ + x, \frac{1}{2} + y, z \\ y, z \\ - x, y - \frac{1}{2}, z \\ y, z - 1 \end{array}$	5 5 5 5 5 5 5 5 5	5(1 <sup>ii</sup> ) 5(2 <sup>ii</sup> ) 5(2 <sup>ii</sup> ) 5(2 <sup>ii</sup> ) 5(2 <sup>ii</sup> ) 5(3 <sup>i</sup> )	$\bar{x}, y, 1-z$ $\frac{1}{2} - x, y + \frac{1}{2}, x, y, z - 1$ $\bar{x}, y, 1-z$ $\frac{1}{2} - x, y - \frac{1}{2}, x, y, z$	l — z l — z	S(3 <sup>ii</sup> ) S(4 <sup>i</sup> ) S(4 <sup>iii</sup> ) S(4 <sup>iii</sup> ) S(4 <sup>iv</sup> ) S(4 <sup>v</sup> )			
$P^{i}-Cr(1^{i})$ $-Cr(2^{i})$ $-Cr(1^{ii})$ $Cr(1^{i})-S_{\epsilon} oc$	3.844 ( 3.815 ( 3.018 (	[17] P <sup>i</sup> —C [17] Cr(2 <sup>i</sup> [15]	r(2 <sup>ii</sup> ) )–Cr(1 <sup>i</sup> ) –C(1 <sup>ii</sup> )	3.052 3.696 3.558	(15) (2) (1)	Si Si Si Si Si Si	$\begin{array}{c} (3^{i}) - S(3^{ii}) \\ (3^{i,i}) - S(4^{i,i}) \\ (3^{i,i}) - S(2^{i,i}) \\ (3^{i,i}) - S(2^{i,i}) \\ (3^{i,i}) - S(2^{i,i}) \\ (4^{i}) - S(2^{i,i}) \\ (4^{i,i}) - S(2^{i,i}) \end{array}$		3 · 2 3 · 6 3 · 6 3 · 3 3 · 2 3 · 2	57 (34) 36 (05) 15 (07) 33 (11) 98 (31) 18 (04)
Cr(1 <sup>i</sup>	)- $S(1^{i})[-S(1^{ii})]$ - $S(3^{i})[-S(3^{ii})]$ - $S(4^{iii})[-S(4^{ii})]$	 )] '')]	2 · 380 (23 2 · 488 (11 2 · 411 (11	) ) )		Si Si Si	$(4^{i,ii}) - S(2^{i,ii}) - (2^{i,ii}) - S(2^{i,ii}) - (2^{i,ii}) - S(4^{i,ii}) - S(4$	-S(3 <sup>i,ii</sup> ) -S(3 <sup>i,ii</sup> )	3.4 63. 63.	99 (16) 98 (09) 32 (09)
S(3 <sup>i</sup> )- S(1 <sup>i</sup> )-	$\begin{array}{c} -Cr(1^{i}) - S(1^{i}) \\ -S(4^{iii}) \\ -S(1^{ii}) \\ -S(3^{ii}) \\ -S(4^{iv}) \\ -Cr(1^{i}) - S(4^{iv}) \\ -Cr(2^{i}) - S(4^{iv}) \\ -S(4^{iv}) \\ -$	) )	95.91 (09 96.74 (08 86.33 (09 81.76 (09 170.32 (03 84.31 (09 93.53 (09	) ) ) )		Si Si Si Si Si Si Si Si	$(2^{i,i}) - S(3^{i,i}) - (4^{ii,i}) - S(4^{i,i}) - (4^{ii,i}) - S(2^{i,i}) - (4^{i,i}) - S(2^{i,i}) - (4^{i,i}) - S(2^{i,i}) - (4^{i,i}) - S(2^{i,i}) - (2^{i,i}) - S(4^{i,i}) - (2^{i,i}) - S(3^{i,i}) - S($	$\begin{array}{l} -S(4^{(,i)}) \\ -S(2^{(,i)}) \\ -S(4^{(,i)}) \\ -S(2^{(,i)}) \\ -S(3^{(,i)}) \\ -S(3^{(,i)}) \\ -S(4^{(,i)}) \\ -S(4^{(,i)}) \\ -S(3^{(,i)}) \end{array}$	52- 64- 58- 56- 64- 55- 60- 57-	69 (08) 96 (07) 64 (06) 42 (07) 25 (09) 67 (08) 09 (09) 75 (07)
S(4 <sup>iii</sup> ) S(3 <sup>i</sup> )- S(1 <sup>i,ii</sup>	$-S(1^{ii})$ -Cr(1 <sup>i</sup> )-S(4 <sup>iv</sup> -S(3 <sup>ii</sup> ) )-S(3 <sup>i,ii</sup> )	)	177.05 (14 86.30 (09 3.257 (34) 3.616 (06	) ) )		S S	$(3^{i,ii})$ -S $(3^{i,ii})$ -S $(2^{i,ii})$ -	-S(2 <sup>i, ii</sup> ) -S(3 <sup>ii, i</sup> )	66 - 55 -	52 (07) 73 (06)
S(1 <sup>i,ii</sup> S(1 <sup>i,ii</sup> S(4 <sup>iii</sup> ) S(1 <sup>i,ii</sup> S(3 <sup>i,ii</sup>	)-S(3 <sup>ii,i</sup> ) )-S(4 <sup>iii,iv</sup> ) )-S(4 <sup>iv</sup> ) )-S(4 <sup>iv,iii</sup> ) )-S(4 <sup>iii,iv</sup> )		3 · 331 (11) 3 · 216 (04) 3 · 298 (31) 3 · 491 (16) 3 · 663 (05)	, ) ) )		$P^{i}$ -S₄ tetrah $P^{i}$ -S(3 <sup>i</sup> ) -S(1 <sup>iii</sup> ) S(3)	nedron $2 \cdot 089 (04)$ $2 \cdot 014 (08)$ $3^{i} - P^{i} - S(1^{iii})$	pi	$-S(2^{iii})$ $-S(4^{v})$ 108	2.019 (08) 2.088 (14) 93 (20)
S(3 <sup>i, ii</sup> S(1 <sup>i, ii</sup> S(1 <sup>i, ii</sup> S(1 <sup>ii, i</sup> S(1 <sup>ii, i</sup>	$)-S(1^{i,ii})-S(4^{i})-S(4^{i})-S(4^{i})-S(4^{ii})-S(4^{ii},i^{i})-S(4^{ii},i^{i})-S(3^{i})-S(3^{i},i^{i})-S(3^{i})-S(3^{i},i^{i})-S(3^{$	ii,iv) ii,iv) 3 <sup>i,ii</sup> ) , <sup>ii</sup> )	64 53 (09) 52 43 (08) 63 04 (09) 57 70 (07) 66 56 (07)	) ) )		S(1 S(3	$ \begin{array}{r} -S(2^{iii}) \\ -S(4^{v}) \\ I^{iii}) - P^{i} - S(2^{iii}) \\ -S(4^{v}) \\ 3^{i}) - S(1^{iii}) \end{array} $	,	108 108 114 103	68 (20) 77 (05) 18 (14) 24 (18) 39 (12)
S(3 <sup>i,ii</sup> S(3 <sup>i,ii</sup> S(1 <sup>ii,i</sup> S(3 <sup>i,ii</sup> S(1 <sup>ii,i</sup> S(4 <sup>iv,</sup>	$\begin{array}{l} ) - S(1^{ii,i}) - S(3^{i}) \\ - S(1^{ii,i}) - S(4^{i}) \\ - S(3^{i,ii}) - S(4^{i}) \\ - S(4^{iii,iv}) - S(4^{ii,iv}) \\ - S(4^{iv,iii}) - S(4^{iv,iii}) \\ - S(1^{ii,i}) - S(4^{iv,iii}) \\ - S(1^{ii,i}) - S(4^{iv,iii}) \\ - S(4^{$	i, i) ii, iv) 1 <sup>ii, i</sup> ) 4 <sup>iii, iv</sup> ) 4 <sup>iii, iv</sup> )	55 74 (06 64 90 (09 59 65 (09 63 04 (09 64 79 (07 58 75 (06	) ) ) )		S(1 S(2	$-S(2^{iii}) -S(4^{v})$ $I^{iii})-S(2^{iii}) -S(4^{v})$ $2^{iii})-S(4^{v})$		3 · 3 3 · 5 3 · 8 3 · 2 3 · 2	38 (12) 95 (33) 668 (06) 216 (04) 218 (04)
$S(4^{iv})$ $Cr(2^i)-S_6 oc$	iii)—S(4 <sup>iii, iv</sup> )—S etahedron	6(1 <sup>ii,i</sup> )	56 46 (07	)		S(1 S(4 S(4 S(3		2 <sup>iii</sup> ) 2 <sup>iii</sup> ) 1 <sup>iii</sup> ) <sup>iii</sup> )	63 58 58 58	51 (10) 28 (10) 21 (10) 36 (07)
Cr(2 <sup>i</sup>	)-S( $3^{i}$ )[-S( $3^{ii}$ ) -S( $4^{i}$ )[-S( $4^{ii}$ ) -S( $2^{i}$ )[-S( $2^{ii}$ )	)  )] )]	2 · 438 (12 2 · 440 (11 2 · 396 (23	) ) )		S(4 S(4 S(3 S(3 S(4	$S^{i} = S(3^{i}) - S(2^{iii}) - S(3^{i}) - S(3^{i}) - S(3^{i}) - S(3^{i}) - S(1^{iii}) - S(1^{iiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiiii}) - S(1^{iiiii}) - S(1^{iiiii}) - S(1^{iiiiiii}) - S(1^{iiiiiii}) - S(1^{iiiiiii}) - S(1^{iiiiiiii}) - S(1^{iiiiiiiii}) - S(1^{iiiiiiiiiiii}) - S(1^{iiiiiiiiiii}) - S(1^{iiiiiiiiiiiiii}) - S(1^{iiiiiiiiiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiii) - S(1^{iiiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiiii) - S(1^{iiiiiiiiii) - S(1^{iiiiiiiiiiiiii) - S(1^{iiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiii) - S(1^{iiiiiiiiiiiiiiiiiiiiiiiiiiiiiiiiiiii$	3 <sup>i</sup> ) <sup>iii</sup> ) <sup>(v</sup> ) <sup>iii</sup> )	55 66 58 66	47 (07) 47 (07) 39 (07) 49 (07)
S(3 <sup>i</sup> )	Cr(2 <sup>i</sup> )S(4 <sup>i</sup> ) S(2 <sup>i</sup> ) S(2 <sup>ii</sup> ) S(3 <sup>ii</sup> ) S(4 <sup>ii</sup> )		96 · 36 (09 96 · 79 (09 87 · 17 (09 83 · 82 (10 170 · 54 (03	) ) ) )		S(1 S(1 S(3 S(1	$(1)^{(1)} - S(3)^{(1)} - S(2)^{(1)} - S(2)$	2011) 2011) 3 <sup>1</sup> )	60- 59- 59-	95 (10) 52 (10) 54 (10)
S(4 <sup>i</sup> ) S(2 <sup>i</sup> )	$\begin{array}{c} -Cr(2^{i})-S(2^{i})\\ -S(2^{ii})\\ -S(4^{ii})\\ -Cr(2^{i})-S(2^{ii})\end{array}$	)	83 41 (09 92 67 (09 85 02 (09 174 69 (15	) ) )						



Fig. 2. Atomic arrangement of the  $CrPS_4$  crystal structure viewed along [010]; the slip plane is indicated.

Compound	Space group Z	Lattice constants	Transformation	Close-packing of S atoms	Coordination of <i>M</i> by S
BPS₄	1222	a = 5.60 Å	$\mathbf{a} = \frac{1}{2}(\frac{\mathbf{a}'}{2} + \mathbf{b}' + \mathbf{c}')$	cubic	tetrahedral
	2	$b = 5 \cdot 25$	$\mathbf{b} = \frac{1}{2}(\frac{\mathbf{a}'}{2} - \mathbf{b}' - \mathbf{c}')$		
		c = 9.04	$\mathbf{c} = \mathbf{c}' - \frac{\mathbf{a}'}{2}$		
AlPS₄	<b>P222</b> 2	$a = 5 \cdot 61 \text{ Å}$ $b = 5 \cdot 67$ $c = 9 \cdot 05$	same as for BPS <sub>4</sub>	cubic	tetrahedral
GaPS₄	$P2_{1}/c$	a = 8.603 Å	$\mathbf{a} = \mathbf{c}' - \frac{\mathbf{a}'}{2}$	hexagonal	tetrahedral
	4	b = 7.778 c = 11.858 $\beta = 135.46^{\circ}$	b = c' c = a' $\beta = 3\beta'/2$		
InPS₄	/4 2	a = 5.60  Å $b = a$ $c = 9.02$	same as for BPS <sub>4</sub>	cubic	tetrahedral
CrPS₄	C2 4	$a = 10 \cdot 871 b = 7 \cdot 254 c = 6 \cdot 140 \beta = 91 \cdot 88^{\circ}$		hexagonal	octahedral

# Table 4. Crystal chemistry of $MPS_4$ thiophosphates



Fig. 3. Schematic illustration of the  $CrPS_4$  crystal structure showing connected coordination polyhedra.

dination polyhedra are comprehensively listed in Table 3. As can be derived from the mean bonding distance of Cr–S (2.426 Å) and P–S (2.053 Å) the bonding of Cr to S is to a great extent ionic (ionic radii: Cr<sup>3+</sup> 0.64, S<sup>2-</sup> 1.84 Å) whereas the mean P–S bond length is considerably shorter than the sum of the ionic radii (P<sup>5+</sup> 0.34 Å). Because of this strong bonding of covalent character (covalent radii: P 1.06, S 1.02 Å) CrPS<sub>4</sub> must be classified as a thiophosphate. Altogether, the S atoms are considerably polarized which is indicated by a mean S–S distance of 3.457 Å.

The Cr atoms are concentrated on plane (001) as is illustrated on Fig. 2, which is viewed along [010]. Again, one octahedron and one tetrahedron are outlined. There is a cation-free region parallel to (001) around  $z = \frac{1}{2}$  explaining the easy slipping mentioned above. The slip plane is indicated on Fig. 2.

Fig. 3 gives a schematic illustration of the  $CrPS_4$  structure in the form of connected coordination polyhedra, only occupied polyhedra being shown (as on all figures the unit cell is outlined by dotted lines). Chains of edge-sharing  $CrS_6$  octahedra run along [010]. Connexion of the chains along [100] is achieved *via*  $PS_4$  tetrahedra. The  $PS_4$  tetrahedron is linked by corner-sharing to a common corner of two  $CrS_6$  octahedra of one chain and by edge-sharing to two  $CrS_6$  octahedra of an adjacent chain.

Apart from the crystal structure of  $BiPS_4$  (Zimmermann *et al.*, 1975), which exhibits a rather strange coordination of Bi by S, and which is not characterized by a close-packed arrangement of S atoms, the crystal structures of all other MPS<sub>4</sub> compounds so far known are geometrically related (Buck & Carpentier, 1973). If  $CrPS_4$  is included into a similar scheme, the geometric relations given in Table 4 are evident. The transformations of the lattice vectors of the Group III thiophosphates into those of  $CrPS_4$  (**a'b'c'**) are shown in the fourth column. Packing type as well as the coordination of *M* by S are also included. In contrast to the  $M^{11}$  thiophosphates incorporated in Table 4 the metal atom of  $CrPS_4$  is octahedrally coordinated by S.

Finally, it should be stated that the arrangement of the S and Cr atoms is practically centrosymmetric. The non-centrosymmetry of the crystal structure is mainly induced by the P atoms. Since the contribution of the P atoms to the total scattering power of the structure is only of about 10% it can be reasoned that the crystal form of  $CrPS_4$  (each crystal face having a parallel counterface) as well as the Wilson plot indicate the pseudo-centrosymmetry mentioned above.

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